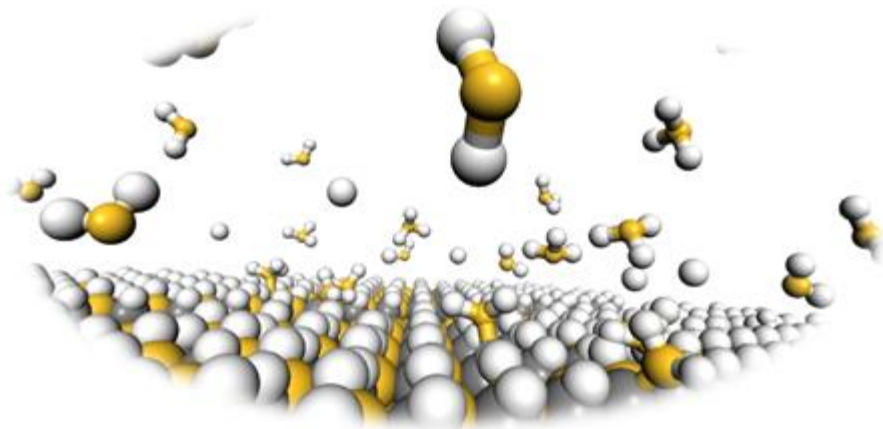


# The difference in nucleation of silicon-based materials during PECVD/PEALD for future S-ALD



Ekaterina Filatova<sup>†</sup>

Dr. Simon Elliott<sup>†</sup>

Dr. Dennis Hausmann<sup>‡</sup>

21.04.2017

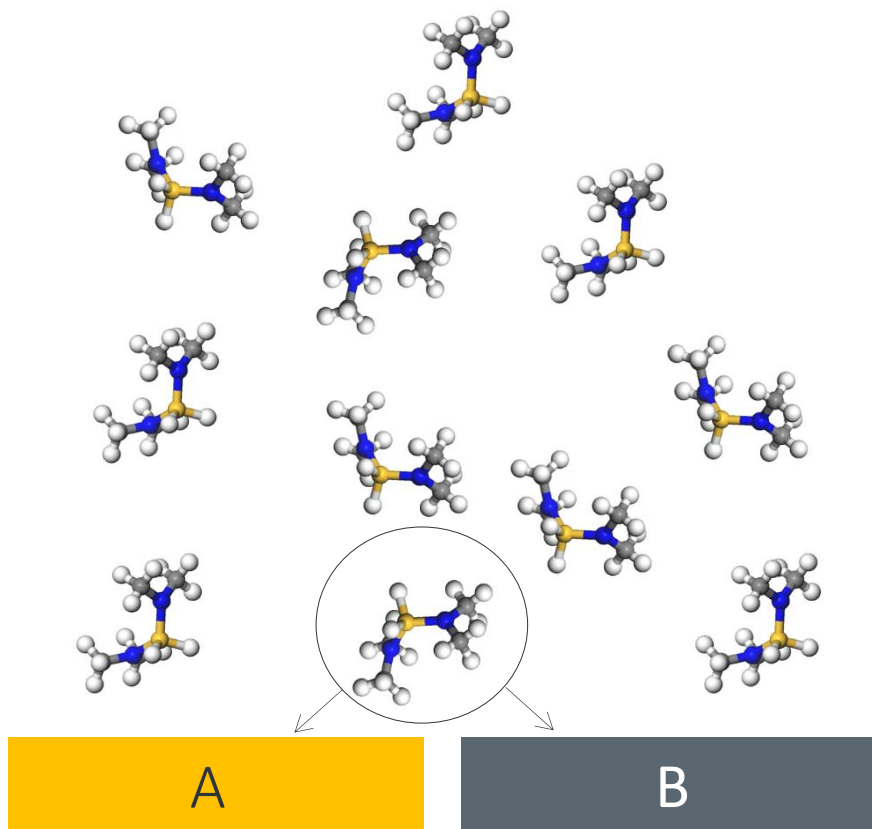
<sup>†</sup> Tyndall National Institute, University College Cork, Ireland

<sup>‡</sup> Lam Research Corporation, Portland, OR, USA

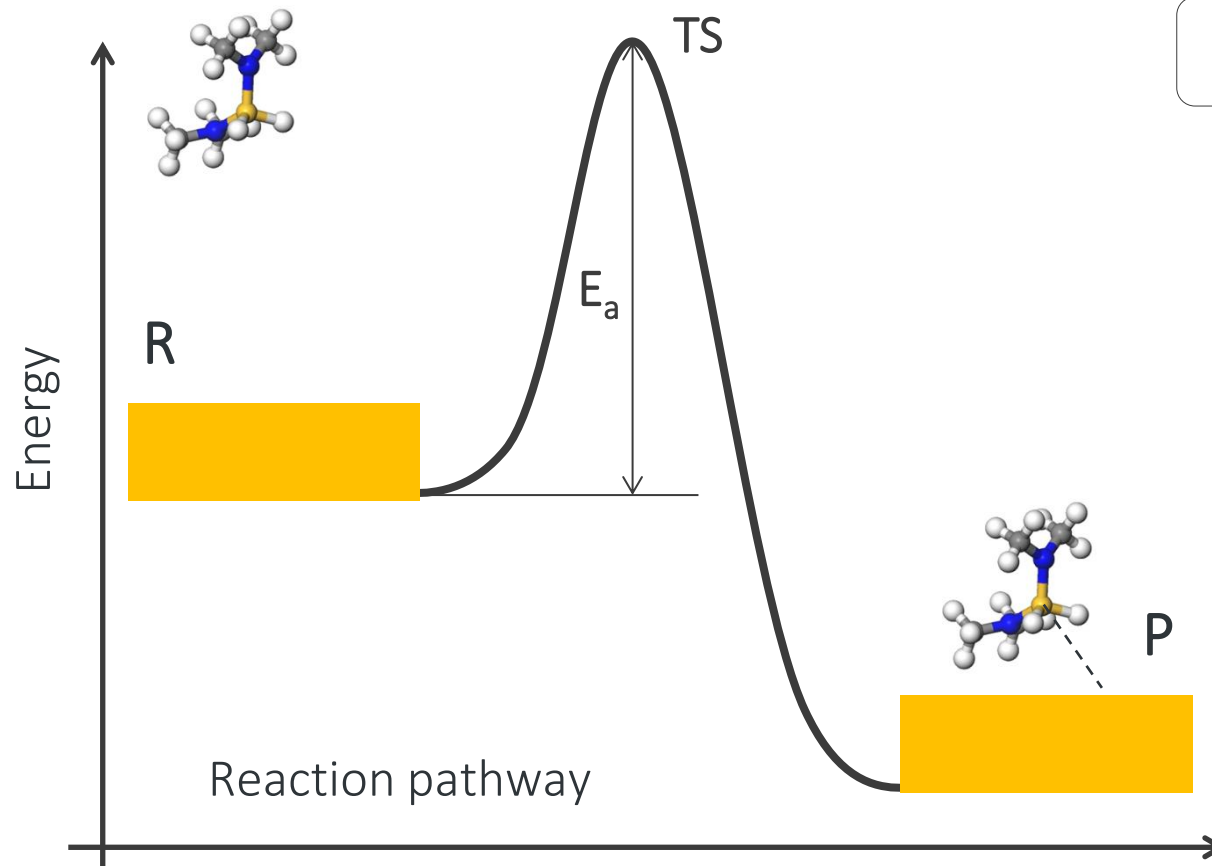


# Theoretical modelling for area selective deposition

Precursor

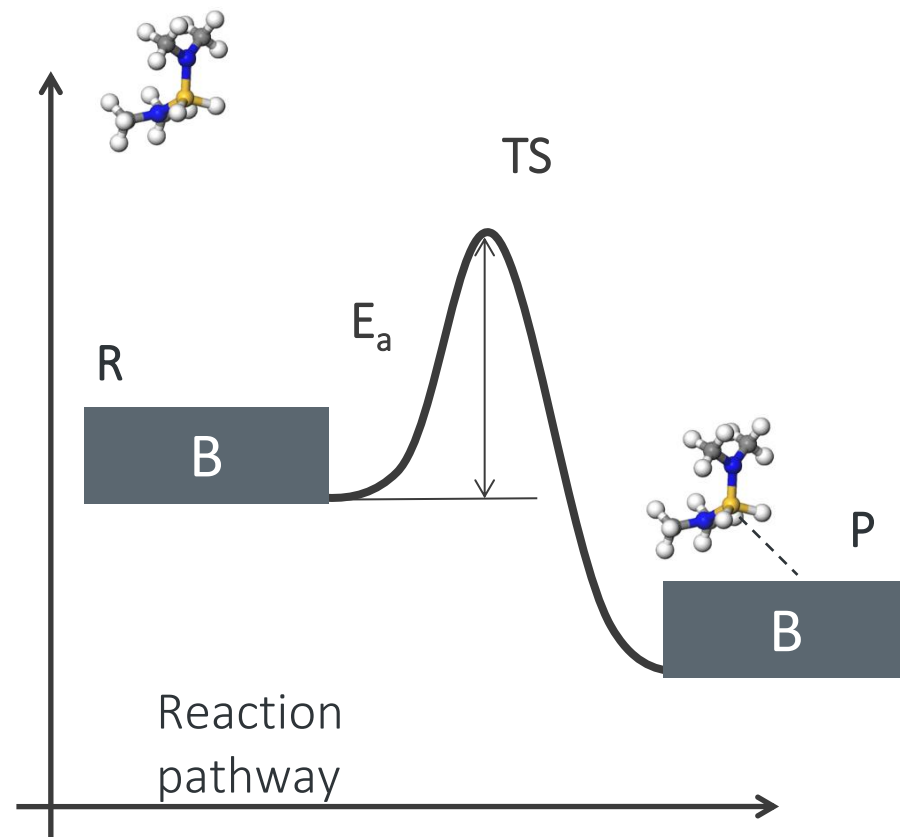
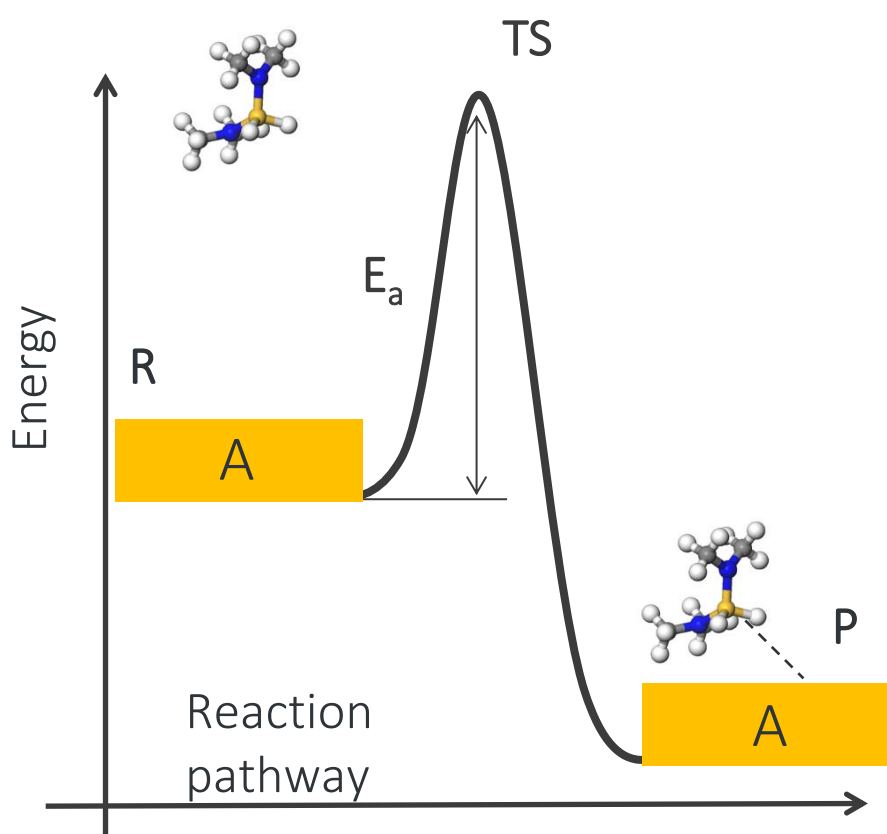


# Theoretical modelling for area selective deposition



R - reactants   TS - transition state   P - products    $E_a$  - activation energy

# Theoretical modelling for area selective deposition



R - reactants   TS - transition state   P - products    $E_a$  - activation energy

# Si-based thin films

SiC

CDO C-SiO<sub>2</sub>

Si

SiO<sub>2</sub>

Si<sub>3</sub>N<sub>4</sub>

- Excellent barrier
- Low-k
- Low wet etch rate

# Si-based thin films

(1)

$\text{SiC}$

(2)

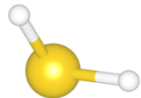
$\text{SiO}_2$

$\text{Si}_3\text{N}_4$

# Silane plasma fragments with H-SiC



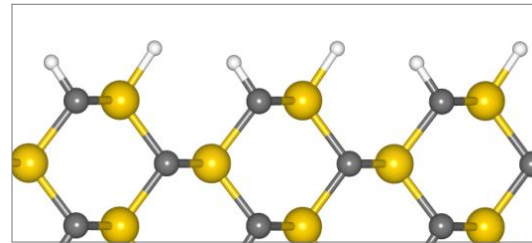
**Silyl** ( $\text{SiH}_3$ )



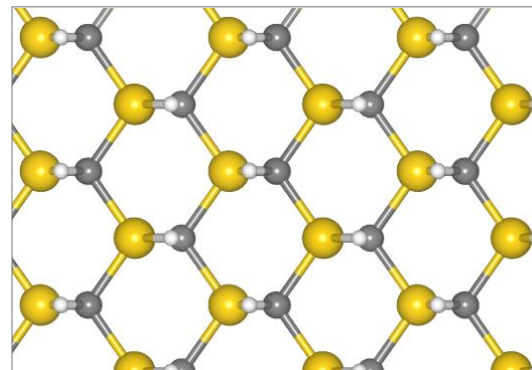
**Silylene** ( $\text{SiH}_2$ )



(011) 3C H-SiC Side view



(011) 3C H-SiC Top view



DFT (VASP)

k-points: 2x2x1

core electrons:

- PAW (projector augmented wave)

valence electrons:

- plane wave basis set with

$E_{\text{cut-off}} = 400 \text{ eV}$

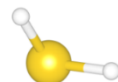
energy barriers:

- CI – NEB

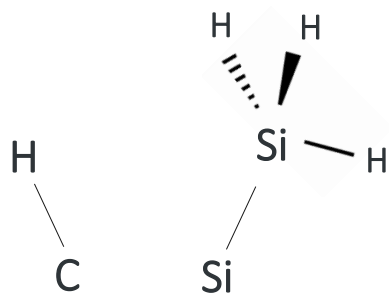
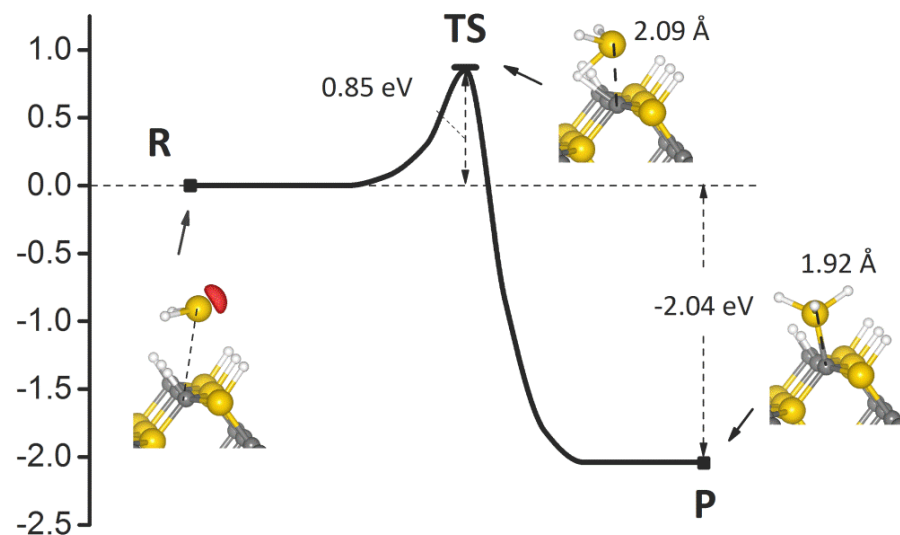
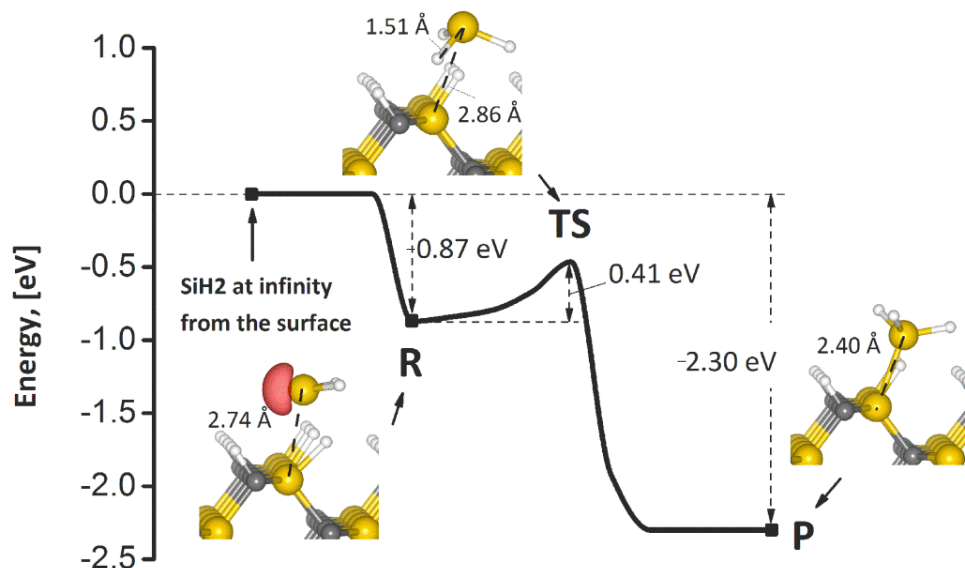
(5 images, force constant 5)

$T = 0 \text{ K}$

# Silane plasma fragment $\text{SiH}_2$ with H-SiC

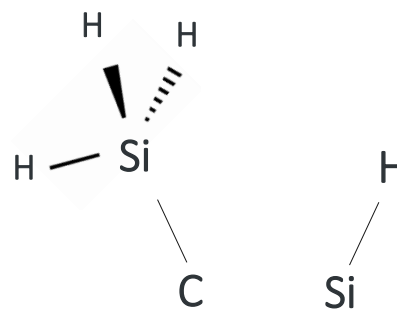


SiC



(0 1 1) H-SiC

Si - H



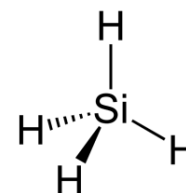
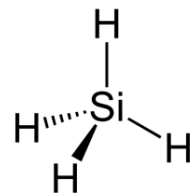
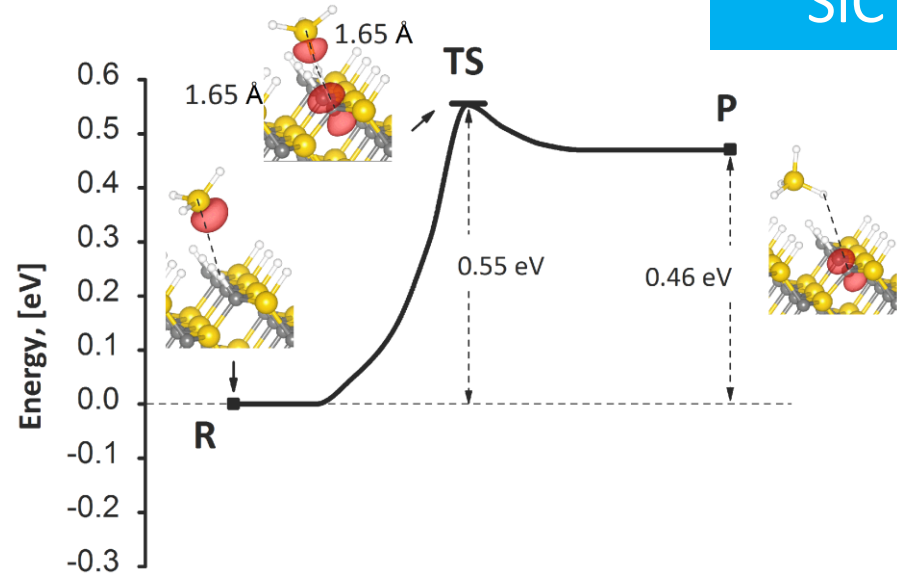
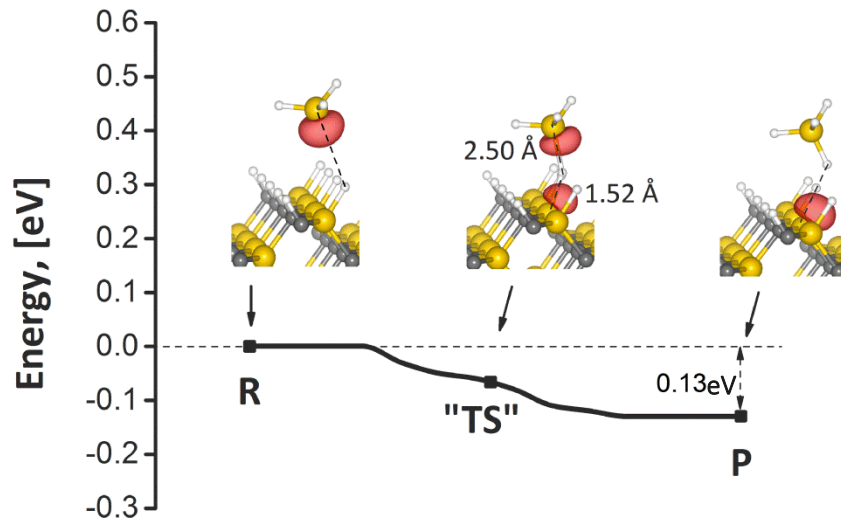
(0 1 1) H-SiC

C - H

# Silane plasma fragment $\text{SiH}_3$ with H-SiC



SiC



(0 1 1) H-SiC

Si-H



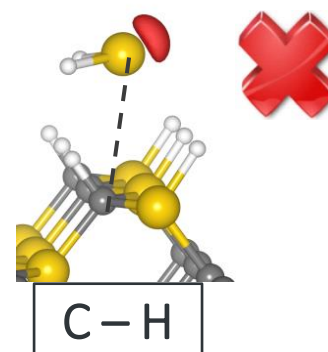
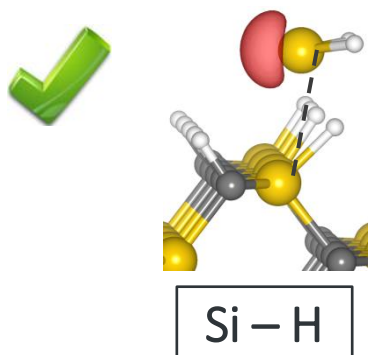
(0 1 1) H-SiC

C-H

# Silane plasma species with H-SiC

## Summary

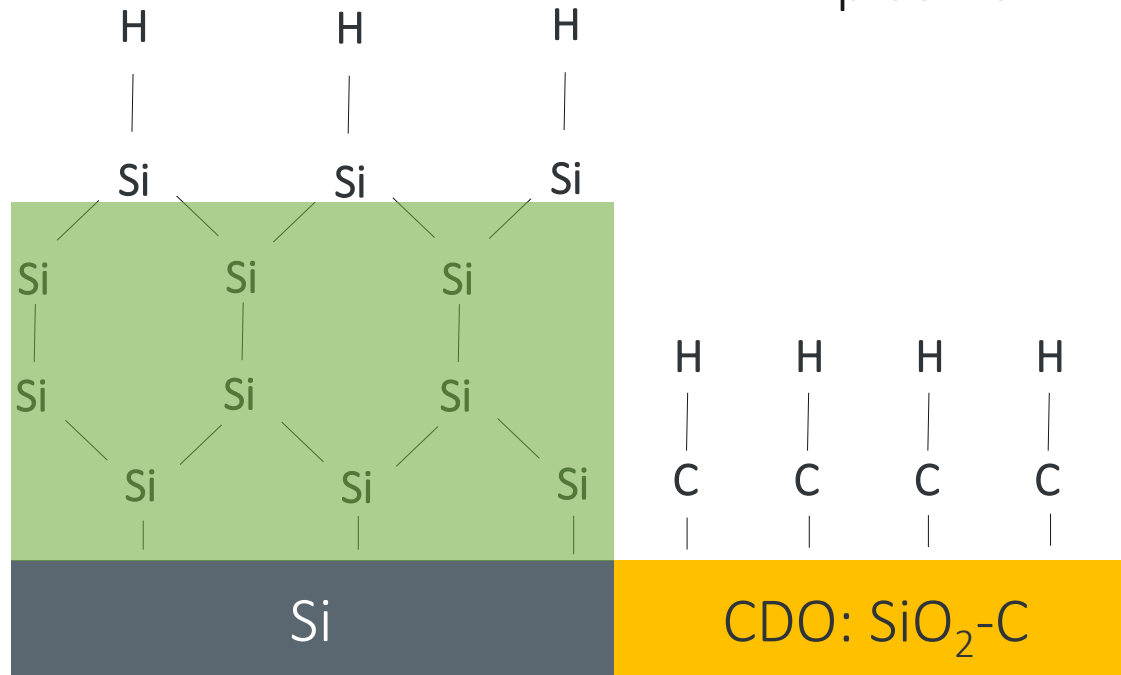
- Both silyl and silylene show selectivity towards insertion into the Si-H bond, rather than the C-H bond



# Silane plasma species with H-SiC

## Summary

Silane ( $\text{SiH}_4$ )  
plasma



SiC

Si

CDO  $\text{SiO}_2\text{-C}$

Selectivity towards Si

# Si-based thin films

(1)

SiC

(2)

SiO<sub>2</sub>

Si<sub>3</sub>N<sub>4</sub>

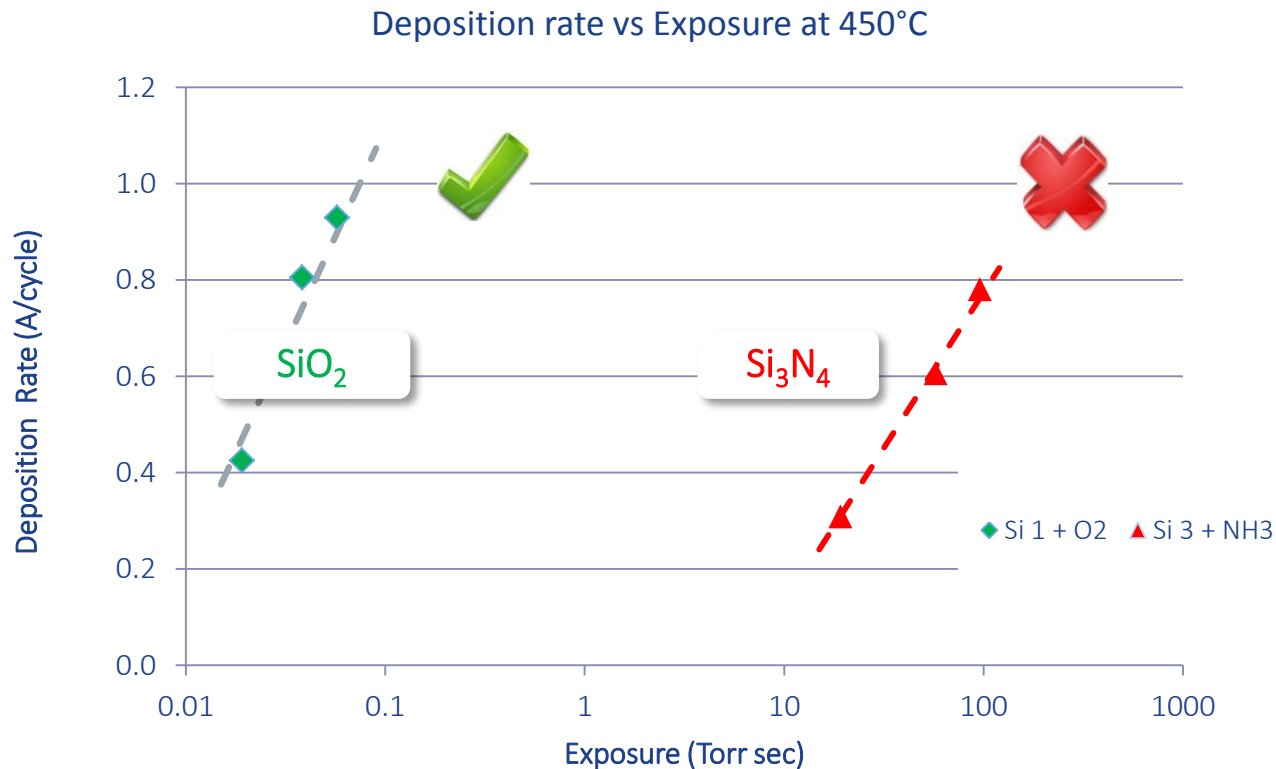
# Reactivity of $\text{Si}_3\text{N}_4$ versus $\text{SiO}_2$

Why is the  $\text{Si}_3\text{N}_4$  reaction slower?

How the problem can be solved?

$\text{SiO}_2$

$\text{Si}_3\text{N}_4$

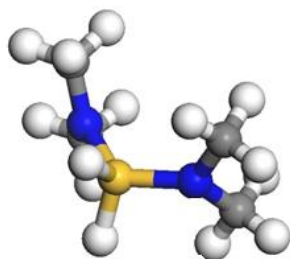


# Reactivity of $\text{Si}_3\text{N}_4$ versus $\text{SiO}_2$

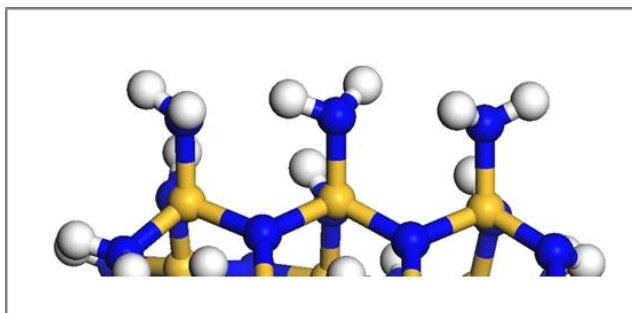
$\text{SiO}_2$

$\text{Si}_3\text{N}_4$

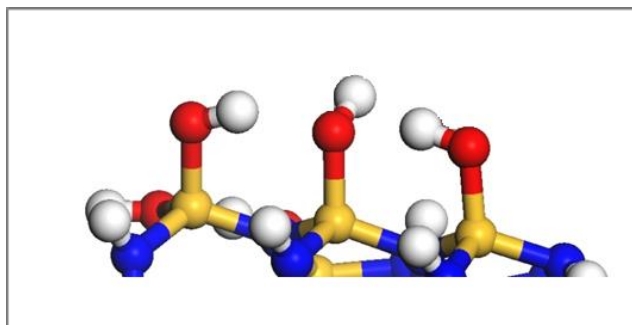
$\text{SiH}_2(\text{NMe}_2)_2$



Side view



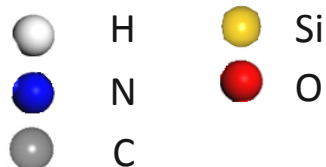
$\text{NH}_2$  surface  
model for  
 $\text{Si}_3\text{N}_4$



$\text{OH}_2$  surface  
model for  
 $\text{SiO}_2$

DFT (TURBOMOLE)

- BP86/TZVPP
- $T = 0\text{K}$

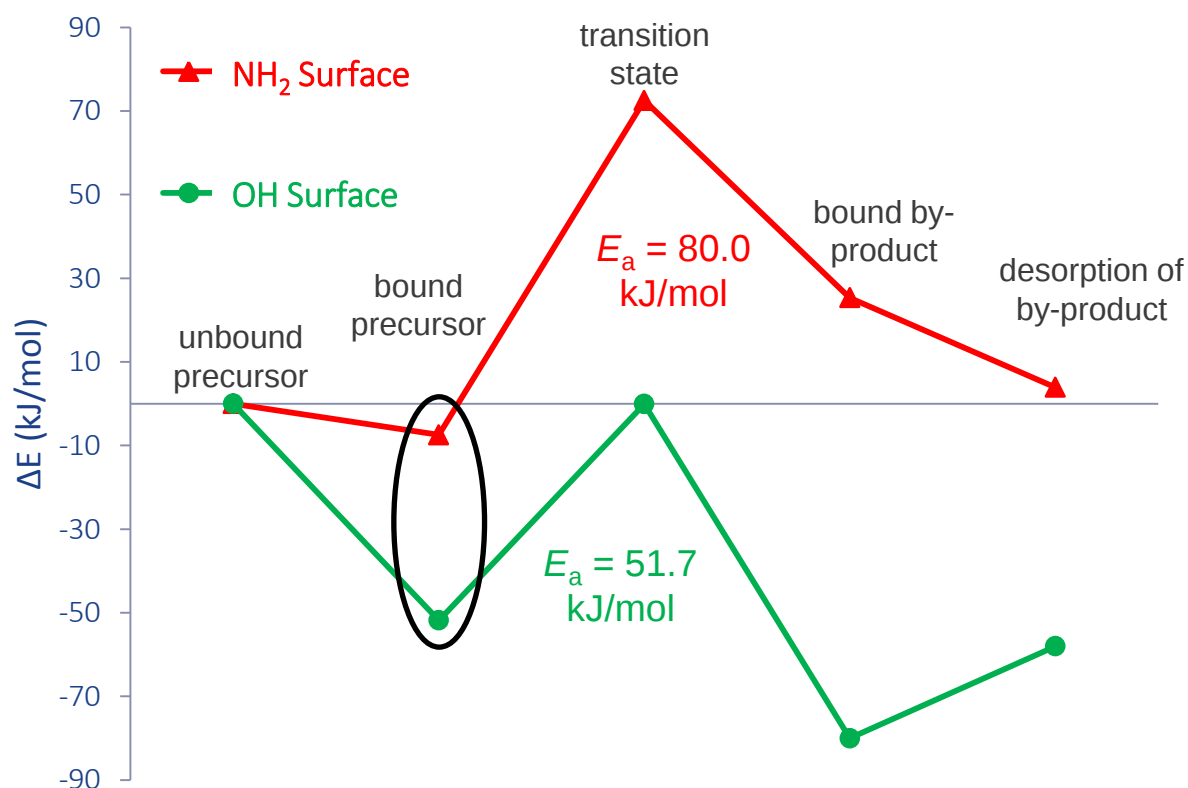
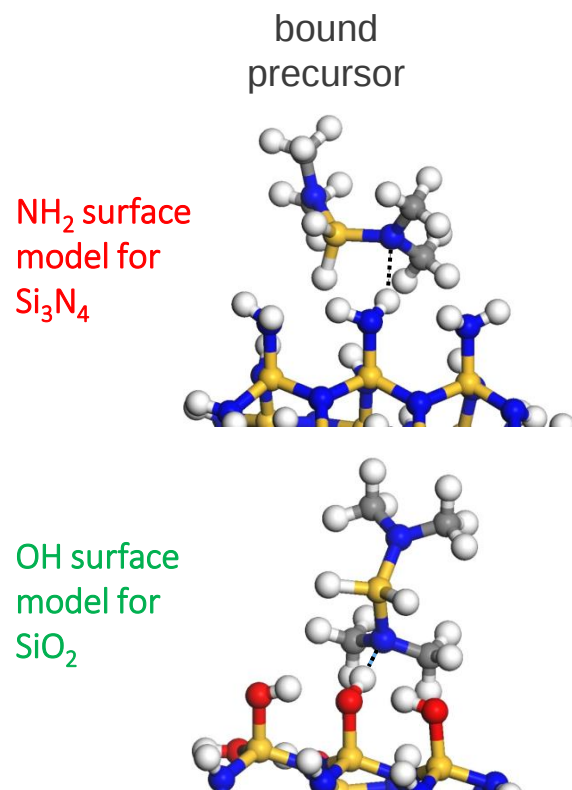


# Reactivity of $\text{Si}_3\text{N}_4$ versus $\text{SiO}_2$

$\text{SiO}_2$

$\text{Si}_3\text{N}_4$

Removing rigid N-H will accelerate ALD reaction

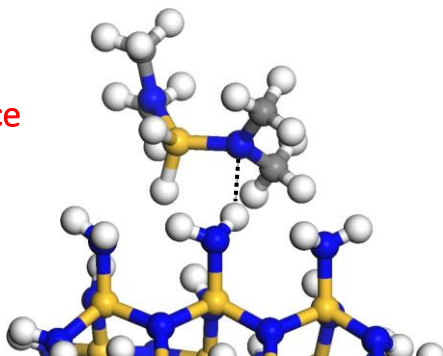


C. M. Murray *et al.*, *ACS Appl. Mater. Interf.* **6**, 10534 (2014)  
C.K. Ande *et al.*, *J. of Phys.Chem.Letters*, **6**, 3610-3614, (2015)

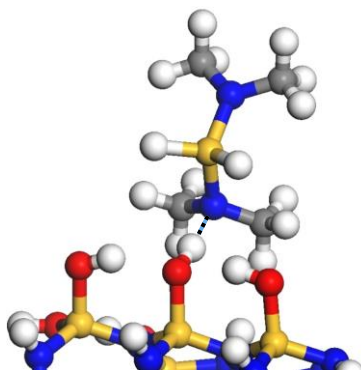
# Reactivity of $\text{Si}_3\text{N}_4$ versus $\text{SiO}_2$

## Summary

$\text{NH}_2$  surface  
model for  
 $\text{Si}_3\text{N}_4$



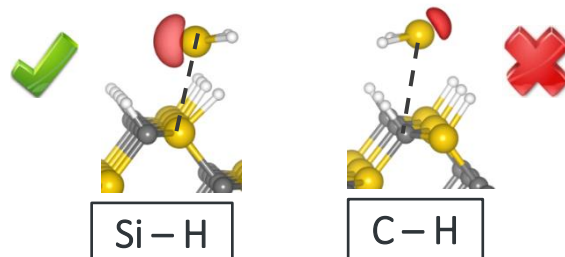
$\text{OH}$  surface  
model for  
 $\text{SiO}_2$



- $\text{Si}_3\text{N}_4$  reaction is slower due to N-H rigidity
- Removing rigid N-H will accelerate ALD reaction

# Silicon-based materials during PECVD/PEALD for future S-ALD

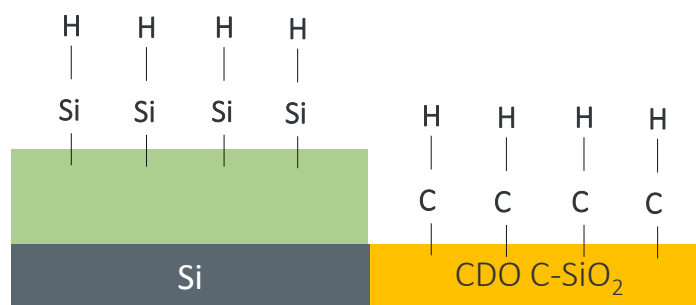
SiC



- Selectivity towards Si-H bond

CDO C-SiO<sub>2</sub>

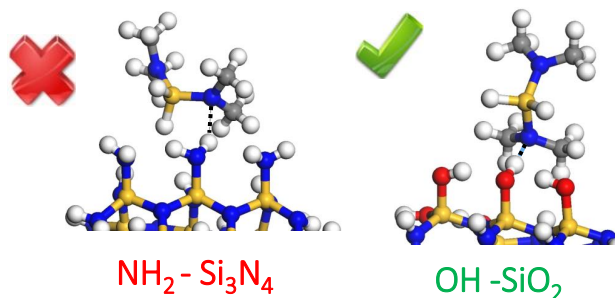
Si



- Selectivity towards Si

SiO<sub>2</sub>

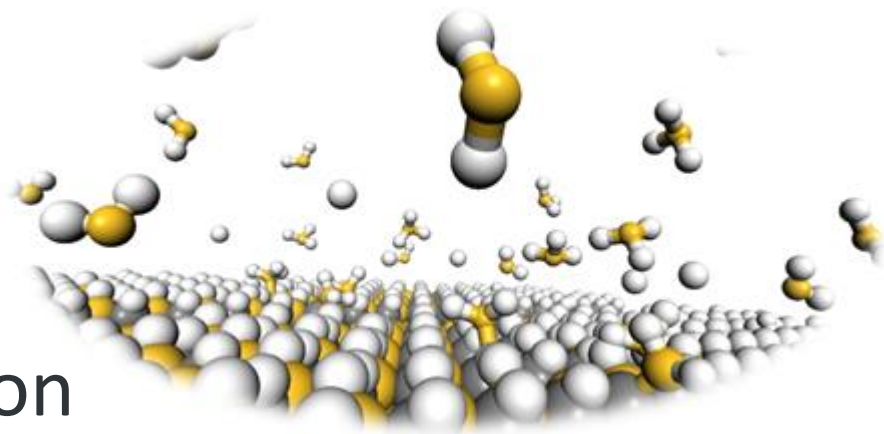
Si<sub>3</sub>N<sub>4</sub>



- Si<sub>3</sub>N<sub>4</sub> reaction slower due to N-H rigidity
- Removing rigid N-H accelerate ALD reaction

# Acknowledgements





# The difference in nucleation of silicon-based materials during PECVD/PEALD for future S-ALD

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